

COALESCENCE THRESHOLD TEMPERATURE IN Ag NANOISLAND GROWTH BY PULSED LASER DEPOSITION

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Abstract. In this study, Ag nanoislands were deposited on glass and Si(100) at substrate temperature from 25 to 450 °C by Pulsed Laser Depositions (PLD). The growth of Ag nanoislands was evidenced from optical absorption surface plasmon resonance (SPR) bands. SPR peaks were not evidently observed in the spectrum of sample deposited at 25 °C, but sample deposited at 150 °C presented a broad SPR peak around $\lambda=870$ nm. SPR peak showed a blue shift to 540-550 nm and became narrower as the substrate temperature increases. Atomic Force Microscope (AFM) showed that by increasing deposition temperature to 350 and 450 °C, Ag islands grow in height and diameter and the number of islands decrease on the surface. Two-probe measurement of electrical resistance of films was also recorded during deposition processes to recognize the coalescence of Ag nanoislands. The results show that the deposition at temperatures below 250 °C leads to a sharp decrease in film resistance while at higher temperatures the islands were separated without resistance decline. To determine coalescence threshold temperature, resistance variation was measured for a cooling substrate, from 300 down to 200 °C, substrate. The result revealed that the coalescence of Ag begins near 230 °C.

1. Introduction

Recently, numerous studies have been focused on the deposition of metal films on different substrates, because the layers interface plays an important role in their wide applications such as catalysts, microelectronic and photovoltaic devices. The growth of atomically flat thin metal films is of great importance both scientifically and technologically. However, the formation of three-dimensional islands on a substrate in the early stages of layer growth (Volmer-Weber growth mode) is very common in thin-film deposition processes. On the other hand, recent discoveries in nanoscience and nanotechnology present opportunity for applications of metal thin-films with a nanostructured morphology such as porous layer, nanoparticles and quantum dots of noble metal deposited on insulator or semiconductor substrates. For example, there is growing interest in utilizing the optical properties of silver nanoparticles as the functional component in various products and sensors. Silver nanostructures being used in numerous technologies and incorporated into a wide array of consumer products that take advantage of their desirable optical, conductive, and antibacterial properties [1-5]. Silver nanoparticles are extraordinarily efficient at absorbing and scattering light and, unlike many dyes and pigments, have a color that depends upon the size and the shape of the particle [6].

Different methods have been used so far for deposition of Ag thin films on different substrates including sputtering [7, 8], evaporation [9], chemical vapor deposition (CVD) [10] and pulsed laser deposition (PLD) [11-13]. PLD presents some advantages over other

methods for the fabrication of metal nanoparticles and the manipulation of their properties, since it is possible to change various parameters such as: laser wavelength, pulse duration, ambient gas pressure, energy per pulse, target-substrate distance and substrate temperature to control the size and distribution of nanoparticles [11]. Nanosecond pulsed laser irradiation of the target surface indicates a complex sequence of events which occurs both during and after impinging of the laser pulse. These include laser absorption at the surface, phase change and mass expulsion in both liquid and vapor states, plasma formation leading to partial absorption in the plume, accelerating of the vapor to hypersonic velocities followed by rapid condensation and finally collapse of the plume onto the substrate surface [14]. Relaxation of plume species upon the substrate that is held at different temperature gives a deposited film involving morphology with a strongly temperature dependency. Deposition temperature-dependent morphological change in the growth of thin Ag films was previously observed [15-17].

Ex-situ measurements of film resistance as a function of thickness requires growing a series of samples of varying film thickness. In contrast, by continuously *in-situ* monitoring the resistance of a growing film, a complete thickness dependence curve can be obtained during growth of a single sample. The *in-situ* method thus makes it practical to vary growth parameters, such as substrate temperature, gas partial pressure and deposition rate, and optimize processes based on the resulting resistance curves [18]. We reported previously the oxidation dynamics of growing tungsten oxide film in vacuum and O₂ partial pressure via resistance measurement during growth by pulsed laser deposition [19]. The purpose of this study is to investigate the effects of substrate temperature on the growth, coalescence and resistance of thin Ag films.

2. Experimental

The nanostructured Ag films were prepared using a KrF excimer laser (248 nm, 20 ns, 10 Hz) to ablate a rotating Ag target in a vacuum chamber at a pressure of 3×10^{-5} Torr. The laser beam was imaged onto the target, at an incident angle of 45°. Films were deposited on Si(100) and glass substrates placed 9.5 cm directly in front of the target. The crystal structure of samples were analyzed, using X-ray diffractometer (Philips EXPERT MPD) with Cu- α ($\lambda = 0.154$ nm) radiation. Optical transmission spectra of films on glass were measured with a dual beam UV-vis spectrophotometer (model PerkinElmer lambda 25). Surface topography of deposits on silicon were measured by atomic force microscope (AFM). Films' electrical resistance was recorded during deposition by two-probe method. In order to make good electrical contact with the growing film, thick Ag contact pads were deposited onto the 5 mm \times 10 mm glass sheets by PLD. The resistance was recorded once per second using digital multimeter model ESCORT 176.

3. Results and discussion

Typical XRD pattern of a pulsed laser deposited Ag film at 450 °C on Si(100) substrate is shown in Fig.1. Based on XRD measurements all the deposited Ag films display fcc structure.

In Fig. 2 optical extinction ($\text{Ln}(T^{-1})$) spectra of Ag/glass deposited at temperatures, from 25 °C to 450 °C, are presented. The absorption peak due to Surface Plasmon Resonance (SPR) is not observed in the spectrum of sample deposited at 25 °C and it be considered as a uniform continues film because the shoulder of the curve in the near IR region is the well-known characteristic of continues high reflective Ag films. Sample deposited at 150 °C involves a broad SPR peak over the curve shoulder around $\lambda=870\text{nm}$.

When substrate temperature increases form room temperature to 150 °C, the corresponding shoulder bends a little at the end region of the spectrum (around 1100 nm) and the corresponding SPR peak is forming. The resonance peak has a blue shift to 540-550 nm

and become narrower as the substrate temperature increases to 250, 350 and 450 °C. The low extinction at high wavelength region arises from increasing in transmittance of the films because it will be shown by AFM images that the higher substrate temperature gives a structural open layer in which light will be able to transmit through the particle separating space. Generally, there is a correlation of the peak position, red shift and widening of the SPR with the increase of the mean diameter of nanoparticles and widening of size distribution [11, 20, 21].

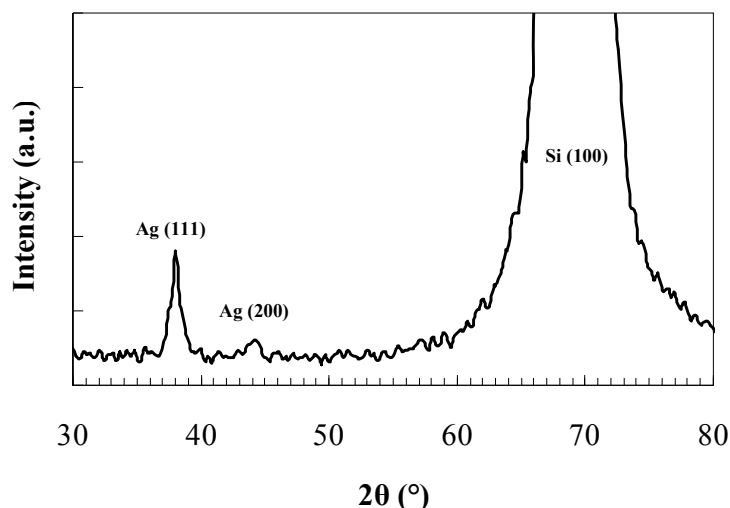


Fig. 1. XRD pattern of Ag/Si(100) deposited by PLD at 450 °C.

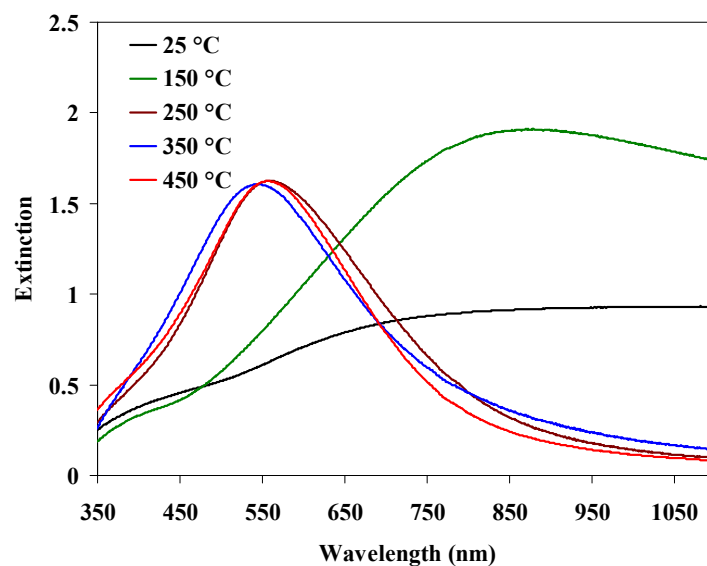


Fig. 2. Optical extinction of Ag/glass deposited at different substrate temperatures, from 25 to 450 °C. XRD pattern of Ag/Si(100) deposited by PLD at 450 °C.

In order to have a three-dimensional (3D) view of the Ag nanoparticles and to measure their height the films were analyzed by AFM. 3D AFM images and corresponding z-height distribution histogram of Ag/Si(100) deposited at different substrate temperatures are shown in Fig. 3. As figure shows, sample deposited at 150 °C indicates a continuous surface including some Ag particulate deposited from plume specimens. Sample deposited at 250 °C mainly involves many small islands of Ag almost uniformly distributed. By increasing

deposition temperature to 350 and 450 °C, these Ag islands grow in height and diameter, and at the same time, the number of island seems to decrease among the surface. The RMS surface roughness obtained from AFM images are 0.3, 2.9, 3.0, and 3.4 nm for 150, 250, 350, and 450 °C, respectively. The films deposited at 350 and 450 °C are basically formed by spherical and ellipsoidal Ag nanoparticles, meanwhile deposits at lower temperatures (250 °C) shows smaller particles and that deposited at 150 °C has a percolated morphology. Morphological evolution from AFM data along with the SPR blue shift occurring with increase in the substrate temperature, indicate that the Ag nanoislands become separate of each other when they are growing at higher temperatures. Therefore percolating of Ag nanoislands is expected to be observed below a temperature limit. In situ electrical resistance measurement is a capable tool to monitor the percolation threshold for layers with a percolating growth regime.

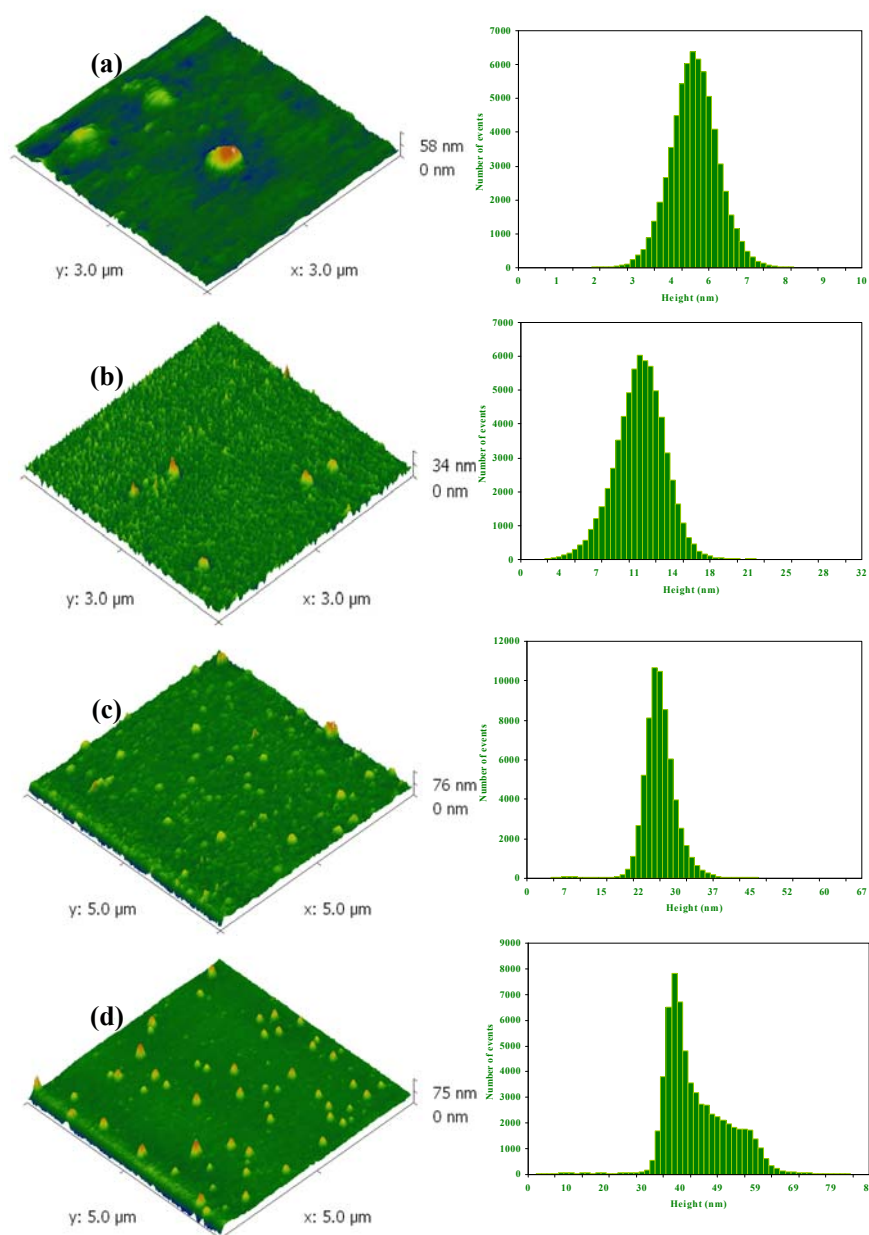


Fig. 3. AFM images and corresponding z-height distribution histogram of Ag/Si(100) deposited at temperatures of (a) 150, (b) 250, (c) 350, and (d) 450 °C.

In the following, electrical resistances of growing film are studied for films growth both on substrate with constant temperatures and on a substrate that is cooling during deposition process. The first one is used for measurement percolation thickness (or the number of laser shots required for percolation) and the other one for recognizing a percolation threshold temperature limit below which obtaining a contiguous percolated film will be possible. Figure 4 shows the Ag/glass electrical resistance R versus the numbers of incident pulses measured in situ during growth at different temperatures of 25, 150, 250, 350, and 450 °C. Furthermore, a different sample was initially deposited at 450 °C then it was allowed to cool down to 25 °C, then deposition continued over the prior one by which an Ag 25 °C /Ag 450 °C /glass structure was achieved.

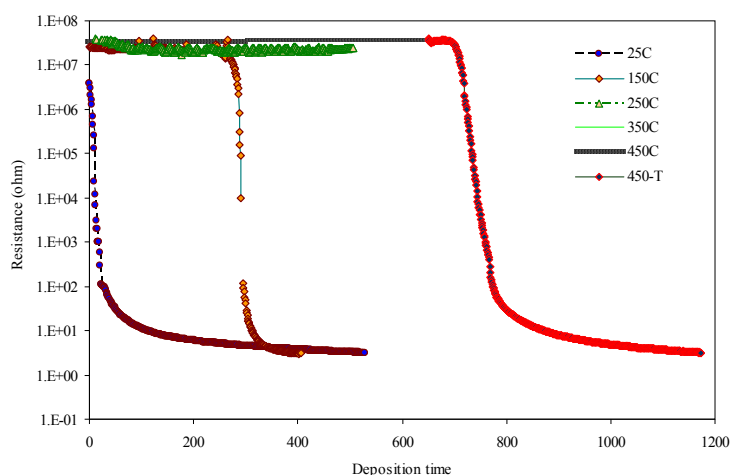


Fig. 4. Time variation of Ag/glass resistance at different substrate temperatures.

As figure shows, the lowest temperature film, 25 °C, presents a sharp reduction in resistance as the laser illumination begins, then it gradually drops to several ohms by further laser illumination. The drop in resistance is the result of Ag nanoislands coalescence which gives formation of a conducting network. Based on other works, it is expected that when the metal film reaches percolation, a precipitous drop in the resistance will be observed [22]. A coalescence threshold is defined as the thickness where the first conducting link forms across the surface and the electrical resistance decreases rapidly as the film starts coalescing [23]. It should be noted that the nearly constant value before any drop in resistance is observed, originates from plume-substrate contact hence the points measured before coalescence cannot be interpreted as film resistance. The resistance of sample deposited at 150 °C has a similar reduction curve but it appears after illuminating ~2500 pulses, and before, it remains almost constant. In contrast, deposition above 250 °C does not represent any reduction in resistance even after 6000 pulses. Temperature dependence of island coalescence is more evidenced by the resistance variation of Ag 25 °C /Ag 450 °C /glass sample. In this case, after delivering ~7000 pulses at 450 °C no coalescence is observed while after cooling the sample down to 25 °C, the sharp decrease in resistance is observable but with a little delayed after the beginning of pulse illumination. According to these results, deposition of Ag above a special temperature limit results in growing films containing disconnects islands. It can be attributed to the increased migration length of silver atoms and the decreased saturated island density at high temperature. To confirm such a temperature dependence effect and find a temperature limit below which coalescence can occur, the resistance variation of a typical sample was monitored during deposition on a substrate with a temperature-cooling rate from 300 down to 210 °C (Fig. 5).

As can be seen, the electrical resistance of growing Ag film during period of cooling remains constant above 230 °C even after 7700 pulses, as is predicable from the previous resistance curves, while below this temperature, it gradually decreases and 1270 pulses after that, when temperature reaches from 230 to 220 °C, the sudden decrease in resistance can be observed. This sharp decline of resistance is due to the reduced substrate temperature and partially to increased Ag film thickness. When the deposition continues, the coalescence between the separated islands happens in a smaller thickness for low substrate temperatures. Regarding to gradual decreasing of resistance at 230 °C it can be concluded that percolation of Ag islands is performed below this temperature.

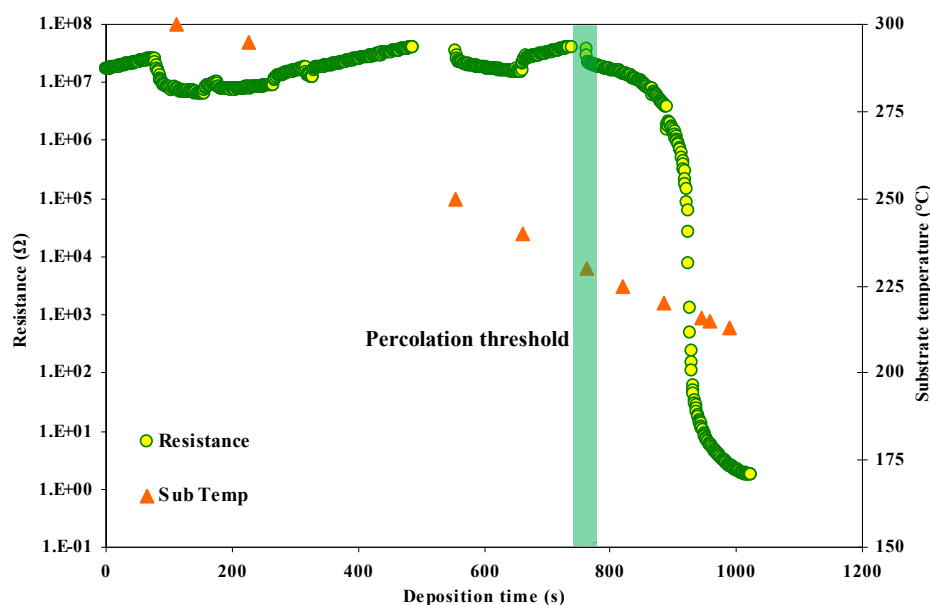


Fig. 5. Resistance variation during deposition Ag/glass during cooling period of substrate from 300 down to 210 °C.

4. Conclusions

In this study, Ag films were grown on Si(100) and glass substrates by PLD at different temperatures.

It was revealed that produced films involve Ag nanoislands. The islands become highly separate at higher temperatures. We found a coalescence threshold temperature for Ag islands around 230 °C.

References

- [1] M. Rai, A. Yadav, A. Gade // *Biotechnology Advances* **27** (1) (2009) 76.
- [2] R. M. Tilaki, A. Irajizad, S. M. Mahdavi // *Applied Physics A* **84** (1-2) (2006) 215.
- [3] G. Yang, Y. Zhou, H. Long, Y. Li, Y. Yang // *Thin Solid Films* **515** (20-21) (2007) 7926.
- [4] G. Yang, W. Wang, Y. Zhou, H. Lu, Z. Chen // *Applied Physics Letters* **81** (21) (2002) 3969.
- [5] G. Yang, D. Guan, W. Wang, W. Wu, Z. Chen // *Optical Materials* **25** (4) (2004) 439.
- [6] B. Wiley, Y. Sun, B. Mayers, Y. Xia // *Chemistry - A European Journal* **11** (2) (2005) 454.
- [7] F.L. Yang, R.E. Somekh, A.L. Greer // *Thin Solid Films* **322** (1-2) (1998) 46.
- [8] W.M. Kim, D.Y. Ku, K.S. Lee, B. Cheong // *Applied Surface Science* **257** (4) (2010) 1331.

- [9] T. Petkova, V. Ilcheva, P. Petkov, G. Socol, C. Ristoscu, F. Sima, C.N. Mihailescu, I.N. Mihailescu, C. Popov, V. Boev, J.P. Reithmaier // *Journal of Optoelectronics and Advanced Materials* **12 (3)** (2010) 650.
- [10] Saizhen Wei, Zusui Mao, Jie Yu, Yongming Zheng, Xiaofeng Chen, Zhenkong Kexue yu Jishu Xuebao // *Vacuum Science and Technology* **17 (6)** (1997) 428.
- [11] J.C. Alonso, R. Diamant, P. Castillo, M.C. Acosta-García, N. Batina, E. Haro-Poniatowski // *Applied Surface Science* **255 (9)** (2009) 4933.
- [12] E. Fazio, F. Neri, P. M. Ossi, N. Santo, S. Trusso // *Laser and Particle Beams* **27 (2)** (2009) 281.
- [13] K. Sturm, S. Fähler, H.U. Krebs // *Applied Surface Science* **154** (2000) 462.
- [14] M.S. Tillack, D.W. Blair, S.S. Harilal // *Nanotechnology* **15 (3)** (2004) 390.
- [15] M. Miyazaki, H. Hirayama // *Surface Science* **602 (1)** (2008) 276.
- [16] T.C. Zhang, Z.X. Mei, Y. Guo, Q.K. Xue, X.L. Du // *Journal of Physics D* **42 (6)** (2009) 065303.
- [17] J.M. Warrender, M.J. Aziz // *Physical Review B* **75 (8)** (2007) 085433.
- [18] K.B. Gylfason, A.S. Ingason, J.S. Agustsson, S. Olafsson, K. Johnsen, J.T. Gudmundsson // *Thin Solid Films* **515 (2)** (2006) 583.
- [19] M. Ranjbar, A. Iraj Zad, S.M. Mahdavi // *Applied Physics A: Materials Science and Processing* **92 (3)** (2008) 627.
- [20] Peng Zhou, Hai-Yang You, Jian-Hu Jia, Jing Li, Tao Han, Song-You Wang, Rong-Jun Zhang, Yu-Xiang Zheng, Liang-Yao Chen // *Thin Solid Films* **455-456** (2004) 605.
- [21] T. Donnelly, B. Doggett, J.G. Lunney // *Applied Surface Science* **252 (13 SPEC. ISS.)** (2006) 4445.
- [22] S.K. So, H.H. Fong, C.F. Yeung, N.H. Cheung // *Applied Physics Letters* **77 (8)** (2000) 1099.
- [23] J.S. Agustsson, U.B. Arnalds, A.S. Ingason, K.B. Gylfason, K. Johnsen, S. Olafsson, J.T. Gudmundsson // *Applied Surface Science* **254 (22)** (2008) 7356.